

1 **Experimental investigation of the effects of**  
2 **aggregate size distribution on the fracture**  
3 **behaviour of high strength concrete**

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17 **Abstract:** This paper examines the influence of different aggregate size distributions on the  
18 fracture behaviour of high strength concrete. Three-point bend test was performed on 63 notched  
19 beams casted using three aggregate size distributions and two water to binder ratios. The total  
20 fracture energy,  $G_F$ , and critical stress intensity factor,  $K_{IC}$ , were used to determine the fracture  
21 characteristic of concrete. The results show that the values of  $G_F$  decrease substantially with  
22 increasing coarseness of aggregate grain structure,  $\lambda$ . Values of  $K_{IC}$  also decreased but  
23 demonstrated only limited dependence on  $\lambda$ . In contrast, reducing the total w/b ratio substantially  
24 increases the value of  $K_{IC}$  but had no measurable effect on  $G_F$ .

25 **Keywords** *Aggregate size distribution, Aggregate grading, Fracture energy,*  
26 *Stress intensity factors, size effect;*

27

28 **Highlights:**

- 29 • Aggregate size distribution affects the fracture behavior of high strength  
30 concrete  
31 • Fracture energy and stress intensity factors measured for 63 notched beams

- 32 • Strong dependence of fracture energy upon aggregate size distribution.
- 33 • Limited dependence of stress intensity factor upon aggregate size
- 34 distribution.
- 35 • Reducing the water to binder ratio increases the stress intensity factor
- 36 • Reducing the water to binder ratio have no measurable effect on the
- 37 fracture energy.
- 38 • Maximum value of fracture energy for high strength concrete is limited by
- 39 the strength of aggregates used.

## 40 **1 Introduction**

41 Concrete is widely used in the construction of buildings, bridges and other  
42 infrastructure around the world. Its use is driven by its flexibility of form, the  
43 relative simplicity of manufacture and the widespread availability of both binder  
44 materials, such as Portland cement, and inert, graded, granular aggregates. As a  
45 particulate composite material, the physical and mechanical properties of  
46 hardened concrete are strongly influenced by both its constituent materials and the  
47 proportions in which they are combined. The aggregate typically occupies more  
48 than 70% of the volume of a concrete mix and plays an important role in  
49 determining the physical properties and mechanical behaviour of both the fresh  
50 and hardened material (Chen and Liu 2004). The interactions that occur at the  
51 interface between the aggregate particles and the cement paste which surrounds  
52 and binds them influences many of the properties of hardened concrete, including  
53 strength, stiffness and fracture toughness. Historically, normal strength concrete,  
54 with a compressive strength of less than 50 MPa, has been used in the  
55 construction of plain, reinforced and prestressed concrete structures (Comitte  
56 Euro-International du Beton 2010). However, over the past 30 years, there has  
57 been increasing use of, and reliance on, so-called high-strength concrete for the  
58 creation of ultra-high-rise buildings and long-span bridges (Wolinski et al. 1987;

59 American Concrete Institute 2010). High-strength concretes routinely have  
60 compressive strengths in the range 60-90 MPa although much higher values can  
61 be achieved. This requires good mix design, the appropriate selection of  
62 constituent materials combined with the use of specialised admixtures. As a  
63 consequence, there is significant interest in understanding the mechanisms that  
64 govern the failure behaviour of such high-strength ceramic composites.

65

66 Whilst the fracture resistance of Portland cement concrete is known to be  
67 dependent on its compressive strength and water/binder ratio, w/b, other  
68 parameters such as the size, shape, surface texture, and volume fraction of the  
69 aggregate have also been found to influence the fracture mechanics parameters of  
70 hardened concrete (Hillerborg 1985; Wolinski et al. 1987; Mihashi et al. 1991;  
71 Zhou et al. 1995; Wu et al. 2001; Xiao et al. 2004; Elices and Rocco 2008; Ince  
72 and Alyamac 2008; Königsberger et al. 2014). In normal strength concrete both  
73 the fracture toughness, i.e. the critical stress intensity factor,  $K_{IC}$ , and the fracture  
74 energy,  $G_F$ , of concrete have been found to increase with the compressive strength  
75 of the concrete (Ince and Alyamac 2008), and the maximum aggregate size  
76 (Hillerborg 1985; Wolinski 1987; Mihashi 1991) and volume fraction of the  
77 aggregate (Amparano and Roh 2000). Similar behavior have also been reported  
78 for the self compacted concrete (Karamloo et al. 2016). However, aggregate type,  
79 and associated particle shape and surface texture has been found to have little  
80 effect on the facture energy of normal strength concrete (Wu et al. 2001; Xiao et  
81 al. 2004).

82

83 The compressive strength of high strength concrete also affects the measured  
84 values of both  $K_{IC}$  and  $G_F$  (Wu et al. 2001; Xiao et al. 2004), which increases as

85 the maximum aggregate size increases (Chen and Liu 2004). Unlike normal  
86 strength concrete, the values of  $G_F$  for high-strength concrete have been found to  
87 be sensitive to the aggregate type (Zhou et al. 1995; Xiao et al. 2004). Other  
88 results for high-strength concrete show that  $K_{IC}$  increases as the w/b ratio  
89 decreases (Ince and Alyamac 2008). Thus, the w/b ratio and properties of the  
90 aggregate both control the compressive strength and influence the fracture  
91 behaviour of high-strength concrete. Hence, for simplicity, compressive strength  
92 is widely used as an input parameter for calculating  $G_F$  along with the maximum  
93 aggregate diameter (Committee Euro-International du Beton 1993; Bazant and  
94 Becq-Giraudon 2002). However, the results of Chen and Liu (2004), and Ackay et  
95 al. (2012), indicate that both  $K_{IC}$  and  $G_F$  of normal and high-strength concretes  
96 are influenced by the volume fraction of coarse aggregate. This is because the  
97 dissipated fracture energy is dependent on the physical characteristics of the crack  
98 path through the concrete. A tortuous crack path, with associated micro-cracking,  
99 crack-bridging and aggregate interlocking will result in the absorption of more  
100 energy than a smooth crack path. The path of a crack during fracture will be  
101 influenced by the coarseness of internal random grain structure of the concrete,  $\lambda$ ,  
102 which is itself dictated by aggregate grading (Amparano and Roh 2000). This  
103 suggests that aggregate grading can influence the fracture behaviour of concrete.

104

105 In high-strength concrete, the propagation of cracks, whether passing round or  
106 through the aggregates, is dominated by the quality and distribution of the various  
107 size of aggregates (Issa et al. 2003). When using high-quality aggregates, the  
108 aggregate grading (and associated particle-size distribution) is a significant factor  
109 in controlling the coarseness of the internal random grain structure of concrete and  
110 influences the crack path that develops during the fracture process. Chen and Liu

111 (2004), and Ackay et al. (2012) have noted that whilst the volume fraction of  
112 coarse aggregate is thought to influence both  $K_{IC}$  and  $G_F$  of concrete there remains  
113 a lack of experimental evidence in this area. Since the volume fraction of coarse  
114 aggregate reflects the aggregate size distribution, understanding the fracture  
115 behaviour of concrete based on its aggregate size distribution is important.  
116 Therefore, the objectives of this study were to investigate the role of aggregate  
117 size distribution and w/b ratio on the fracture behaviour of high-strength concrete  
118 and resulting values of  $K_{IC}$  and  $G_F$ .

## 119 **2 Experimental setup**

### 120 **2.1 Materials**

121 Flinty river gravel from the Thames Valley river was used for coarse aggregate  
122 with sizes in the range of 5 mm to 16 mm. The fine aggregate was a natural river  
123 gravel of size ranging from 4 mm down to 0.30 mm. A CEM Type I with a  
124 specific surface area of 338 m<sup>2</sup>/kg was employed in the mix combining additional  
125 Pulverized Fuel Ash (Fly Ash) conforming to BS EN450-S category B. A slurry-  
126 based silica fume was also involved in some of the concrete mixes. A super-  
127 plasticiser was employed to allow appropriate workability of the fresh concrete.

128

129 The concrete mix proportions of each material used are shown in Table 1. A total  
130 of six different concrete mixes were produced employing three aggregate grading  
131 curves, designated as 'A', 'B' and 'C', see Figure 1. To achieve the appropriate  
132 grading a batch of the as-received aggregate was sieved into its component  
133 fractions and the required amount of each fraction was then recombined. Two  
134 total water to binder (w/b) ratios of 0.20 (equivalent to a free w/b ratio of  $0.17 \pm$

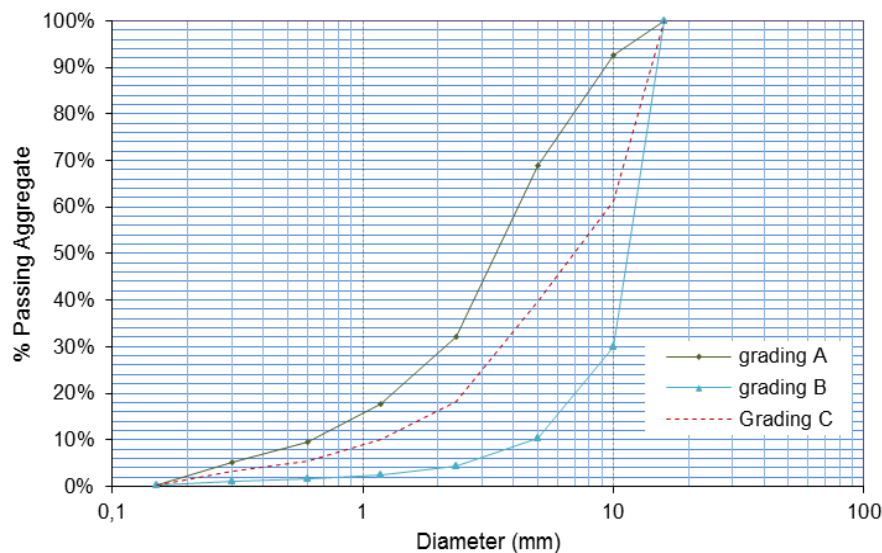
135 0.02) and 0.30 (equivalent to a free water/binder ratio of  $0.23 \pm 0.01$ ) were used,  
 136 and were designated as '1' and '2' respectively, see Table 1. Thus, mix  
 137 'A1' indicates a mix with aggregate grading type 'A' and a total w/b ratio of 0.20.

138 Table 1 Mix proportions of the six concrete mixes tested.

139

Mix	w/b*	Unit weight (kg/m <sup>3</sup> )						
		Aggregate		Cement	PFA	Silica fume	Water	Super-plasticiser
		Coarse	Fine					
A1	0.20	1108	499	547.7	76.7	54.7	135.9	2.64
A2	0.30	1108	499	679	-	-	203.7	1.61
B1	0.20	1552	54	547.7	76.7	54.7	135.9	2.01
B2	0.30	1552	54	679	-	-	203.7	1.42
C1	0.20	1322	284	547.7	76.7	54.7	135.9	2.27
C2	0.30	1322	284	679	-	-	203.7	1.50

140 \*Total water/binder ratio



141

142 **Fig.1** Aggregate grading curves for types 'A', 'B', and 'C'.

143 It is well established that the workability of fresh concrete, as measured by its  
 144 consistence, has a profound effect on the ease of compaction of the material and  
 145 that incomplete compaction can adversely affect the properties of the resulting  
 146 hardened concrete (Neville and Brook 1990). As a consequence, the consistence  
 147 of all the mixes used in this study was kept constant, the target slump value for all  
 148 the mixes was  $120 \pm 20$  mm (equivalent to a consistence class S3 as prescribed by

149 ENV 206, Part 1: 2000). This was achieved by varying the quantity of super-  
150 plasticiser in each mix, Table 1. Since the aggregate to binder ratio is also known  
151 to affect the concrete properties, it is kept constant for all mixes used in this study.

## 152 **2.2 Specimens and Test Setup**

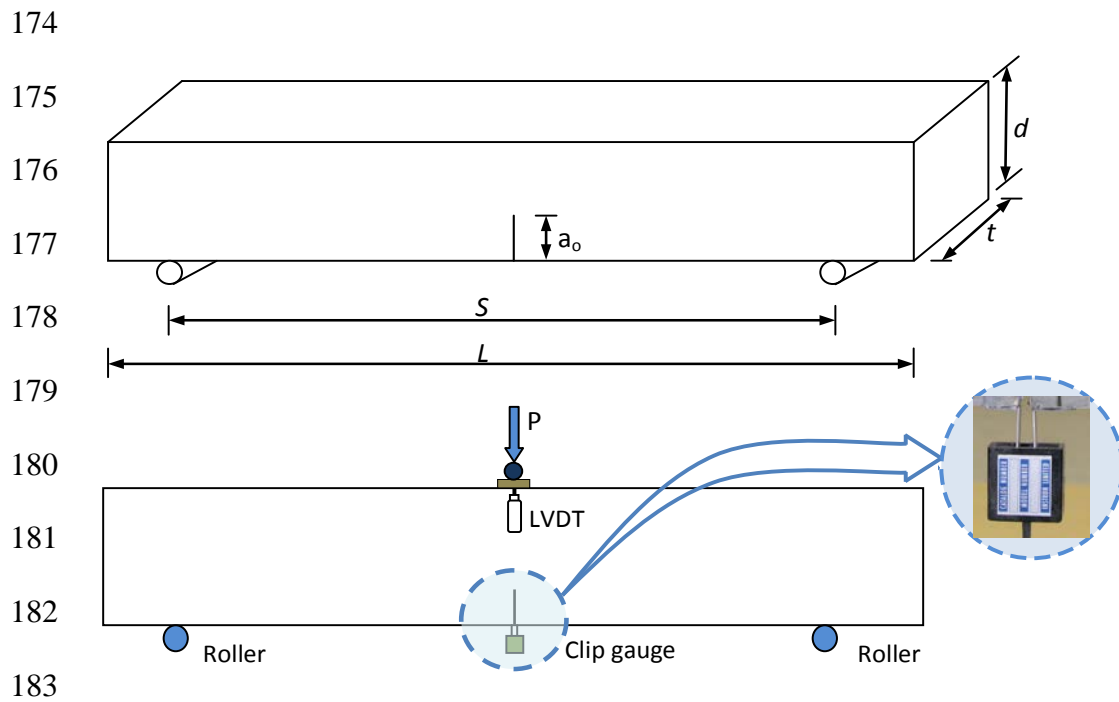
153 At least nine 100 x 100 x 100 mm cubes were tested to determine the average  
154 compressive strength of the hardened concrete, following BS EN 12390: Part  
155 3(2000). The fracture behaviour of the hardened concrete was determined using  
156 100 x 100 x 850 mm beam specimens tested in three-point bend (TPB) following  
157 RILEM TC 50-FCM (1985). All of the specimens were de-moulded  
158 approximately 24 hours after casting and cured under water at  $22 \pm 2$  °C for 30  
159 days prior to testing.

160

161 Figure 2 shows the dimension of the notched beam specimen for the associated  
162 three-point bend test arrangement. The depth (d), the width (t) and the total length  
163 (L) of the beam was 100, 100, and 850 mm respectively with a support span (S) of  
164 800 mm. A notch depth ( $a_0$ ) of 25 mm and width of 2.5 mm was created in all of  
165 the specimens prior to testing.

166

167 The TPB test was carried out using a closed-loop servo-hydraulic testing machine  
168 with a maximum load capacity of 600 kN. The applied load was measured using  
169 a 10 kN load cell. The loading rate was controlled by applying a vertical  
170 displacement of 0.01 mm/s, following the methodology developed by Zhang et al.  
171 (2009). A calibrated linear variable differential transducer (LVDT) was employed  
172 to measure the vertical deflection of the beam at the loading point. A clip gauge  
173 was used to quantify the crack mouth-opening displacement, Figure 2.



184 **Fig. 2** Geometry of specimen and test set up showing detail of clip gauge  
185 attachment

### 186 **3 Concrete properties**

187 The measured values of slump of the fresh concrete and the compressive strength  
188 of the hardened concrete, tested at 30 days, are presented in Table 2. It can be seen  
189 that the average slump of all the mixes were controlled within the target range  
190  $120 \pm 20$  mm.

191

192 The average compressive strength of the concrete mixes with a total w/b ratio of  
193 0.30 (equivalent to a free water/binder ratio of  $0.23 \pm 0.01$ ) ranged from 66-  
194 68 MPa. The average compressive strength of the concrete mixes with a total w/b  
195 ratio of 0.20 (equivalent to a free w/b ratio of  $0.17 \pm 0.02$ ) ranged from 85-  
196 98 MPa. These results are in agreement with the well-known Abrams' law  
197 (Mindess and Young 1983) in that the measured compressive strength increases  
198 with decreasing w/b ratio. Whilst the w/b ratio is the main factor dictating the



199 compressive strength of hardened concrete, local changes in the size, shape and  
 200 distribution of the aggregates within individual test specimens contributes to the  
 201 measured variability. However, this variability is within normal limits for high  
 202 strength concrete mixes.

203

204 Table 2 Slump, mean (30-day) compressive strength and coarseness ratio of concrete mixes

Mix	Slump			Compressive strength			$\lambda$
	mm	N	S	MPa	N	S	
A1	110	4	4.4	85.2	12	5.7	0.32
B1	100	4	6.4	97.6	12	1.4	0.22
C1	115	3	6.3	94.8	9	2.6	0.25
A2	100	4	6.8	68.2	12	3.6	0.32
B2	120	3	5.5	66.0	9	2.9	0.22
C2	130	3	7.8	68.4	9	0.8	0.25

205 N= number of test specimens, S= Standard deviation of sample,  $\lambda$  = coarseness of internal random  
 206 grain structure.

207

## 208 **4 Coarseness of internal random grain structure ( $\lambda$ ),** 209 **and fracture characteristic ( $K_{IC}$ and $G_F$ ) of concrete**

210 The coarseness of the internal random grain structure of the hardened concrete,  $\lambda$ ,  
 211 is presented in Table 2, and the derived fracture characteristics of the concrete, i.e.  
 212  $K_{IC}$  and  $G_F$ , are presented in Table 3 and these are considered in more detail in the  
 213 following sub-sections.

214

#### 215 **4.1. Coarseness of internal random grain structure of concrete**

216 According to Amparano and Roh (2000) the coarseness of the internal random  
217 grain structure of concrete,  $\lambda$ , is defined as:

$$218 \quad \lambda = \frac{1}{D_{ave}(1-V_a)} \quad \text{Eq. (1)}$$

219 where,  $D_{ave}$ , is the average diameter of the aggregate particles, and  $V_a$  is the  
220 volume fraction of the coarse aggregate (above 4 mm) calculated from the mix  
221 design parameters of this study (Table 1) and the grading curves shown in Figure  
222 1. It can be seen from Eq. (1) that the value of  $\lambda$  is dependent on both the volume  
223 fraction and average diameter of coarse aggregate and is considered to be an  
224 important factor affecting the dissipated fracture energy in concrete. The effect of  
225 the aggregate grading used in the concrete mix tested in this study produced  
226 different values of  $\lambda$ , i.e. 'B' - low ( $\lambda = 0.22$ ), 'C' - medium ( $\lambda = 0.25$ ) and, 'A' -  
227 high ( $\lambda = 0.32$ ), see Table 2.

228

#### 229 **4.2 Fracture energy**

230 One parameter that has been widely used to characterise the fracture behaviour of  
231 brittle composite materials such as concrete is the fracture energy,  $G_F$ . In this  
232 study, values of  $G_F$  (for each specimen) were calculated from the corresponding  
233 load vs mid-span deflection curve, Figures 3(a) and 3(c). By following the  
234 recommendations of RILEM TC50-FCM (RILEM 1985) it was possible to  
235 account for the effects of the self-weight of the beam:

$$236 \quad G_F = \frac{W_o + mg\bar{d}_o}{(d - a_o)t} \quad \text{Eq (2)}$$

237 where,  $G_F$  is the total fracture energy,  $W_o$  is the area under the load-deflection  
238 curve,  $m$  is the total mass of specimen between supports,  $g$  is the gravitational  
239 constant,  $\delta_o$  is the deflection at failure,  $d$  is the height of the sample,  $a_o$  is the  
240 depth of the notch and  $t$  is the width of the sample. Values of the mean fracture  
241 energy and standard deviation for the samples tested are presented in Table 3.

242

243 The fracture energy dissipated during the failure process of concrete can be  
244 related to the crack opening, which drives the crack forward. This was monitored  
245 by the load vs crack mouth-opening displacement (CMOD) curves, Figures 3(b)  
246 and 3(d). These were used to assist analysis of the crack-opening behaviour in the  
247 post- peak stress regime.

248

#### 249 **4.3 Critical stress intensity factor ( $K_{IC}$ )**

250 In considering the tensile fracture of concrete it is important to consider the role of  
251 inherent defects, such as cracks, in limiting the stress that can be supported by a  
252 given section of material. In the case of brittle ceramic materials, such as concrete,  
253 the role of cracks can be successfully modeled on the basis of linear elastic  
254 fracture mechanic-LEFM (Shah and Ahmad 1994). This approach assumes that a  
255 suitably oriented crack within the material will start to propagate when the (mode  
256 I) stress intensity factor  $K_I$ , at the crack tip exceeds the critical stress intensity  
257 factor,  $K_{IC}$ , which is also known as the fracture toughness (Zhang and Xu 2011). At  
258 its simplest  $K_I$  can be considered to be a measure of the force tending to cause fast  
259 fracture whilst  $K_{IC}$  is a material property, like stiffness and Poisson's ratio that is a  
260 measure of the materials resistance to crack growth. In a brittle material  
261 containing a crack (of given size) and subject to an externally applied stress fast

262 fracture will not occur provided that  $K_I < K_{IC}$ . Determination of the appropriate  
 263 stress intensity factor is complicated by the fact that  $K_I$  is a function of the crack  
 264 length, the local (and global) geometry, the value and mode of the externally  
 265 applied load and associated boundary conditions (Shah et al. 1995).

266

267 In this study the LEFM approach has been used to analyse the effect of the three  
 268 different aggregate size distributions on the fracture characteristic of high-strength  
 269 concrete tested in three-point bend up to the peak stress. The values of  $K_{IC}$  for the  
 270 individual test samples was carried out following the RILEM TC89-FMT  
 271 (RILEM 1990):

$$272 \quad K_{IC} = \frac{3(P_c + 0.5W) \left[ S(\pi a_c)^{3/2} g\left(\frac{a_c}{d}\right) \right]}{2d^2t} \quad \text{Eq. (3)}$$

273 in which,

$$274 \quad g\left(\frac{a_c}{d}\right) = \frac{1.99 - \left(\frac{a_c}{d}\right)\left(1 - \frac{a_c}{d}\right) \left[ 2.15 - 3.93\left(\frac{a_c}{d}\right) + 2.70\left(\frac{a_c}{d}\right)^2 \right]}{\sqrt{\pi} \left[ 1 + 2\left(\frac{a_c}{d}\right) \right] \left[ 1 - \left(\frac{a_c}{d}\right) \right]^{3/2}} \quad \text{Eq. (4)}$$

275

276 where,  $P_c$  is the critical maximum load,  $W$  is the beam's self-weight,  $S$  is span of  
 277 specimen,  $a_c$  is the critical effective elastic crack length,  $d$  is the specimen depth,  
 278 and  $t$  is the specimen width. The mean, and standard deviation, of the calculated  
 279 values of  $K_{IC}$  are presented in Table 3.

280

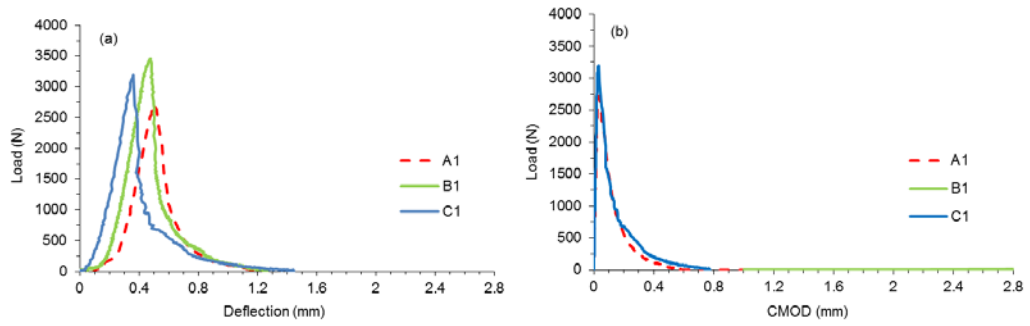
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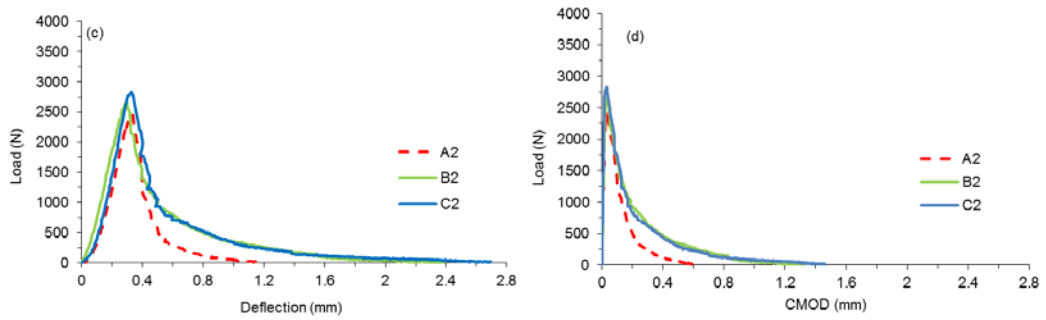
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286



287

288 **Fig. 3** Typical load-deflection and associated load-CMOD curves for concrete  
 289 with;  $w/b = 0.2$  (3(a) & 3(b), and  $w/b = 0.30$  (3(c) & 3(d).

290

291 Table 3 Fracture energy and critical stress intensity factor for concrete mixes.

Mix	w/b ratio	Number of samples	Mean $G_F$	S	Mean $K_{IC}$	S
			N/m	N/m	MPa.m <sup>1/2</sup>	MPa.m <sup>1/2</sup>
A1	0.2	12	93.8	9.1	0.99	0.07
B1	0.2	12	149.9	10.6	1.24	0.11
C1	0.2	9	126.7	14.1	1.26	0.12
A2	0.3	12	85.1	10.4	0.78	0.05
B2	0.3	9	146.5	13.3	0.93	0.11
C2	0.3	9	134.3	11.2	0.71	0.09

292 S= Standard deviation of sample

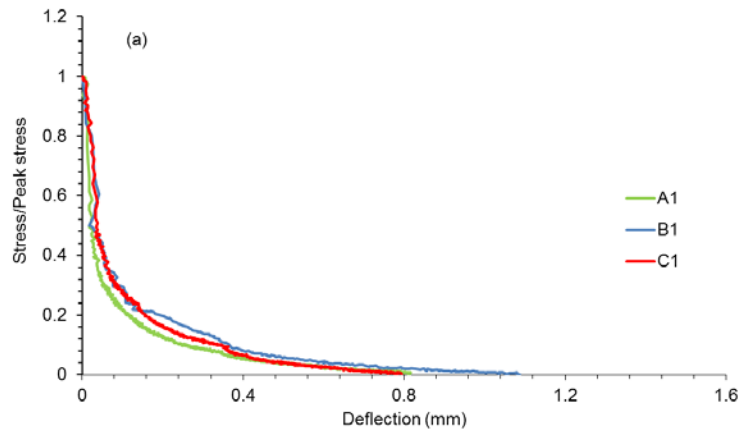
## 293 **5 Discussion**

### 294 **5.1 Effect of aggregate size distribution**

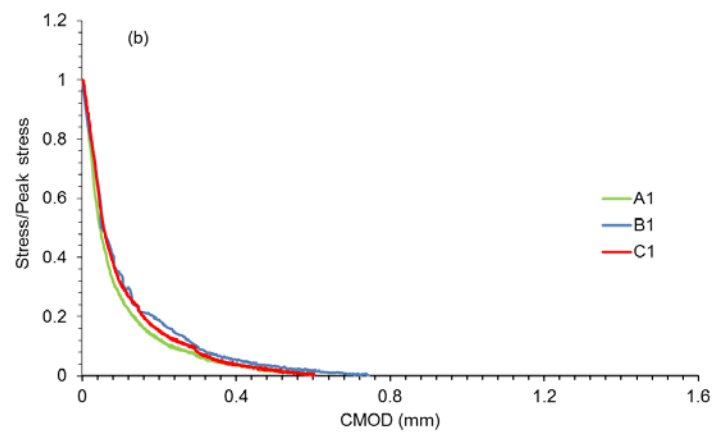
295 According to Hillerborg and Modeer (1976) the pre-peak tensile response of  
296 concrete can best be assessed by examining the stress-strain relationship of the  
297 material whilst the post-peak fracture behaviour, or “softening term”, can be most  
298 accurately assessed by investigating the stress-crack opening curve. In this study  
299 both approaches have been employed to help further elucidate the role of  
300 aggregate size distribution on the fracture behaviour of both normal and high-  
301 strength concrete.

302

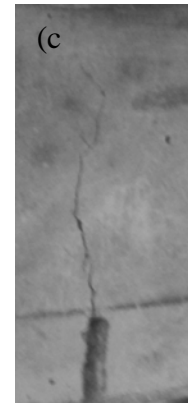
303 The values of  $G_F$  are dependent on the shape and form of the softening curve  
304 making comparisons between samples of the same material and between batches  
305 of different materials complicated. As a consequence, normalised stress  
306 (Stress/Peak Stress) vs deflection curves were used in the analysis of the post-  
307 peak fracture behaviour of the concrete mixes tested. Figure 4(a) compares the  
308 typical deflection and CMOD response obtained for the high-strength concrete  
309 mixes A1, B1 and C1 (w/b ratio = 0.2).



310



311



312 **Fig. 4** Softening response of three point bending test of concrete with w/b ratio of  
 313 0.20 (a) post-peak stress-deflection curve (b) post-peak stress-CMOD curve (c)  
 314 crack path of specimen

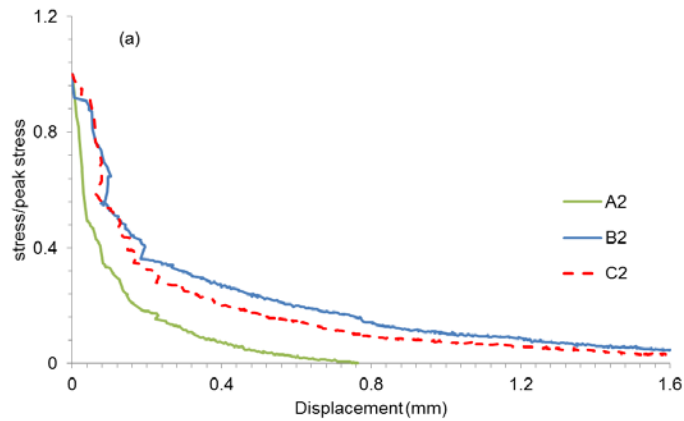
315 It can be seen that there is little difference in the curves obtained for the three  
 316 aggregate gradings suggesting that similar processes are at play although the  
 317 contribution to the total energy absorbed may be different. Indeed it is interesting  
 318 to compare the fracture energy obtained for mixes B1, C1 and A1 (ranked in  
 319 terms of their grain size distribution), which yields the values of  $G_F$  of 149.9,  
 320 126.7 and 93.8 N/m for  $\lambda$ 's of 0.22, 0.25 and 0.32 respectively, see Tables 2 & 3  
 321 and Figure 6(b). Hence for the high-strength concrete, reduction of coarser  
 322 aggregate sizes in the mix (represented by an increase in  $\lambda$  value) is associated  
 323 with a significant reduction in the measured fracture energy. This result is in line

324 with, and confirms, previously published results (Chen and Liu 2004). The values  
325 of  $G_F$  obtained for these mixes may be compared with the measured values of  $K_{IC}$   
326 of 1.24 (B1), 1.26 (C1) , and 0.99 (A1) MPa.m<sup>1/2</sup> respectively for the same  
327 samples, Table 3. This suggests that the critical stress intensity factor to initiate  
328 crack growth is relatively insensitive to the internal random grain structure of  
329 concrete, albeit slightly higher for higher proportions of coarse aggregates in the  
330 mix, Figure 6(a).

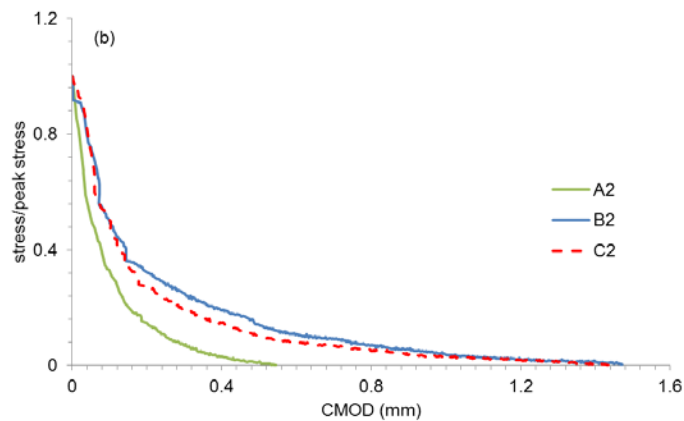
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332 The above results may be compared with those for mixes A2, B2 and C2, Figure  
333 5(a). For these concretes, which have a w/b ratio of 0.3, the normalised curves  
334 obtained demonstrate both a greater variation with the change in  $\lambda$  and a  
335 somewhat slower rate of decay of normalised stress per unit deflection than seen  
336 in Figure 4(a). This suggests that the aggregate size distribution is influencing the  
337 post-peak fracture behaviour in a different way to that of the higher strength  
338 mixes. Again it is interesting to compare the fracture energy obtained for these  
339 mixes when ranked in terms of the grain size distribution, i.e. B2, C2 and A2,  
340 which yields values of  $G_F$  of 146.5, 134.3 and 85.1 N/m respectively. These  
341 values of  $G_F$ , which are in close agreement with those for mixes A1, B1, and C1  
342 (see Table 3), confirm the observation that reducing the coarser grain size in the  
343 mix (increasing  $\lambda$ ) is associated with lower fracture energy. That is to say the  
344 degree of coarseness of internal random grain structure plays a role in the energy  
345 absorption process during the fracture of hardened concrete. As with the higher  
346 strength mixes measured values of  $K_{IC}$  are, when compared to  $G_F$ , relatively  
347 insensitive to changes in  $\lambda$ , see Figure 6(a). The influence of w/b ratio on both  $G_F$   
348 and  $K_{IC}$  is discussed in section 5.2.





349



350

351 **Fig. 5** Softening response of three point bending test of concrete with w/b ratio of  
 352 0.30 (a) post-peak stress-deflection curve (b) post-peak stress-CMOD curve (c)  
 353 crack path of specimen

354

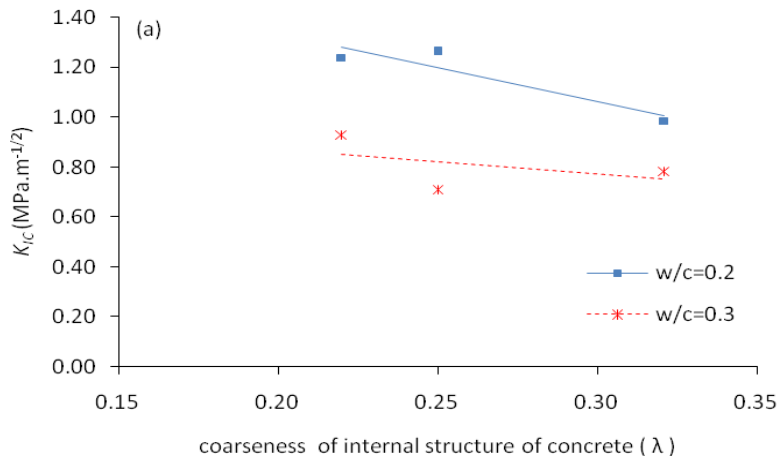
355 Having established the impact of  $\lambda$  on both  $G_F$  and  $K_{IC}$  it is useful to consider  
 356 factors which may be contribute to the overall fracture process. The interfacial  
 357 zone properties and aggregate properties such as size distribution and surface  
 358 texture can play an important role in resisting crack opening. Since the aggregate  
 359 used in these experiments was from a single source, it has been assumed that the  
 360 effect of aggregate surface texture can be safely ignored. Altering the w/b ratio  
 361 has the potential to change the stiffness (and strength) of the interfacial zone that  
 362 exists between the aggregate and the cement matrix. If the strength of the

363 aggregate particles is more than that of the interfacial zone matrix, the aggregate  
364 particles can act as crack growth arresters in the fracture process (Giaccio and  
365 Zerbino 1998). Other toughening mechanisms such as micro-crack shielding and  
366 particle bridging can also assist in the increase of the fracture resistance of  
367 concrete when the aggregate has not failed, i.e. the crack propagates only within  
368 the interfacial zone or through the cement paste. For such a case, the crack path  
369 would be expected to show a strong correlation with  $\lambda$ , i.e. higher coarseness of  
370 internal grain structure of concrete would lead to a more tortuous crack path and  
371 vice versa. This mechanism could explain the longer “tail” in the post-peak stress  
372 vs displacement curve of Figures 5(a) which was obtained with the lower strength  
373 concrete (i.e. higher w/b ratio). In contrast, if the strength of the aggregate is less  
374 than that of the interfacial zone matrix, the crack becomes able to propagate  
375 through the aggregate particles, and is thus less sensitive to  $\lambda$ . Such behaviour  
376 would be expected to produce a relatively straight crack path (Figure 4 (c)) and  
377 smaller “tail” in the post-peak curve such as the one shown in Figure 4(a) for the  
378 higher strength concrete tested here. Hence, the fracture characteristic of high  
379 strength concrete can be assessed on the basis of the fracture parameters and the  
380 slope of the post-peak curves.

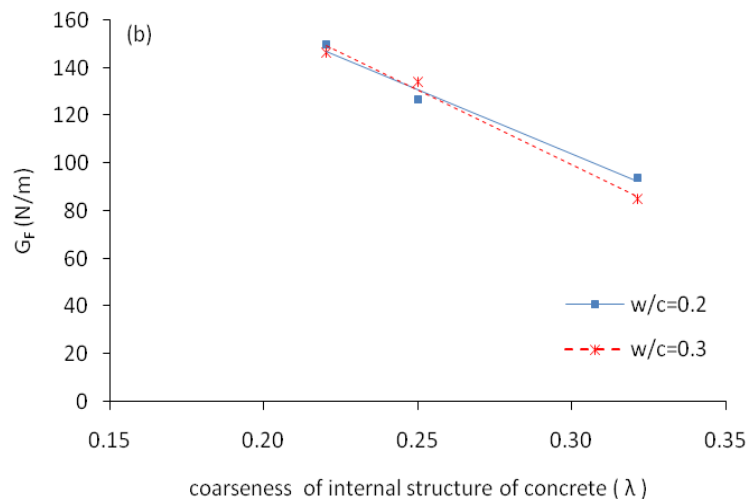
381

382 It should be remembered that a crack will only propagate if the stress intensity at  
383 the crack tip is greater than  $K_{IC}$ . The propagation of a growing crack during a  
384 three-point bend test is related to the crack opening, which can be analysed  
385 through normalised stress versus CMOD curves, Figures 4(b) and 5(b). Although  
386 this is a qualitative analysis, it suggests that whilst internal random grain structure  
387 influences the crack opening in the high w/b ratio concrete (Fig. 5b), its affect at

388 low w/b ratios is small (Fig. 4b). This suggests that the influence of grain size  
 389 distribution on the crack opening decreases with concrete strength, see section 5.2.  
 390



391



392

393 **Fig. 6** The effect of coarseness of internal structure of concrete on (a) the stress  
 394 intensity factor ( $K_{IC}$ ) and, (b) fracture energy ( $G_F$ )

395 It can also be seen from Figure 4 and 5 that the level of ductility varies  
 396 considerably for the beams with different aggregate size distribution and w/b ratios.  
 397 Ductility ratio for the beams is computed to analyze this further and is shown in Table  
 398 4. Ductility ratio is taken as the ratio of the deflection of beam at failure to deflection  
 399 of beam at the yield point (Rao et al., 2010). The Table clearly demonstrates that the

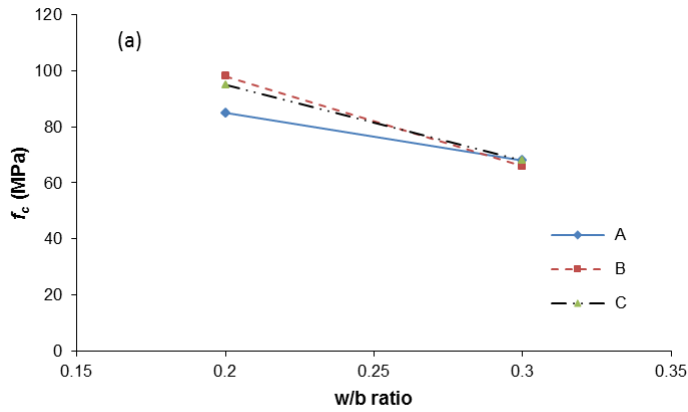
400 aggregate size distribution and w/b ratio significantly influences the level of ductility  
 401 exhibited by high strength concrete.

	Gradation of aggregate					
	A1	B1	C1	A2	B2	C2
yield deflection (mm)	0.5	0.41	0.39	0.33	0.28	0.33
max deflection (mm)	1.20	1.47	1.42	1.17	2.34	2.68
<b>Ductility ratio of beam</b>	<b>2.4</b>	<b>3.6</b>	<b>3.6</b>	<b>3.6</b>	<b>8.4</b>	<b>8.1</b>

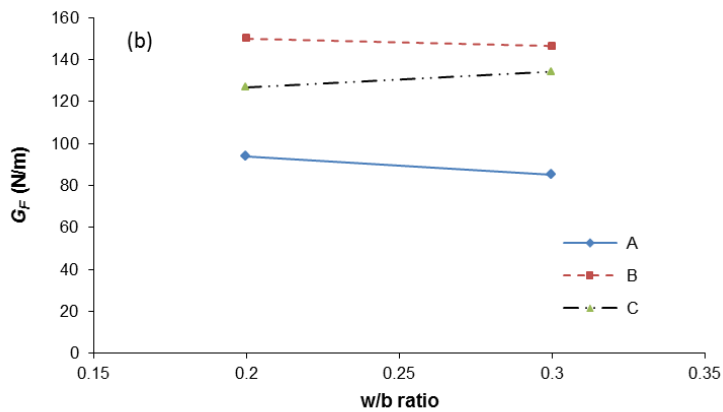
402

403 **5. 2 Effect of water/binder ratio**

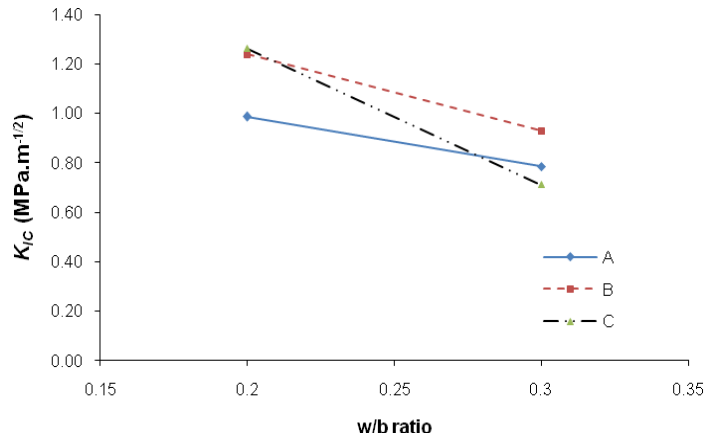
404 Figure 7 shows the effect of w/b on the measured compressive strength,  $f_c$ , critical  
 405 stress intensity factor,  $K_{IC}$ , and fracture energy,  $G_F$  of the six concrete mixes  
 406 tested.



407



408



409

410 **Fig. 7** The effect of w/b ratio on (a) compressive strength, (b) fracture energy,

411 and(c) critical stress intensity factors

412

413 As might be expected  $f_c$  was found to decrease as the w/b ratio increased, Figure

414 7(a). Similarly  $K_{IC}$  sensibly decreased with increasing w/b ratio, Figure7(c). In

415 contrast  $G_F$  was found to be insensitive to w/b ratio, Figure 7(b). This behaviour

416 reflects the result of the hydration process (Neville and Brook 1990) in that the

417 porosity level in the cement paste is related to the amount water in the mix which,

418 in turn, dictates the bond strength between the aggregate and the cement paste.

419 Moreover, supplementary cementitious materials, such as the PFA incorporated in

420 the high strength mixes used here, contribute by reacting with free calcium

421 hydroxide in the cement paste and induces in improvement in the matrix strength

422 (Köksal and Altun 2008). At the same time unhydrated particles of supplementary

423 cementitious materials will, by filling void space in the bulk paste, reduce the

424 porosity of paste and help the formation of a denser aggregate/cement interfacial

425 zone. Where the stiffness of interfacial zone and aggregate are similar, then

426 aggregate grading, which influences the bonding surface area and density of

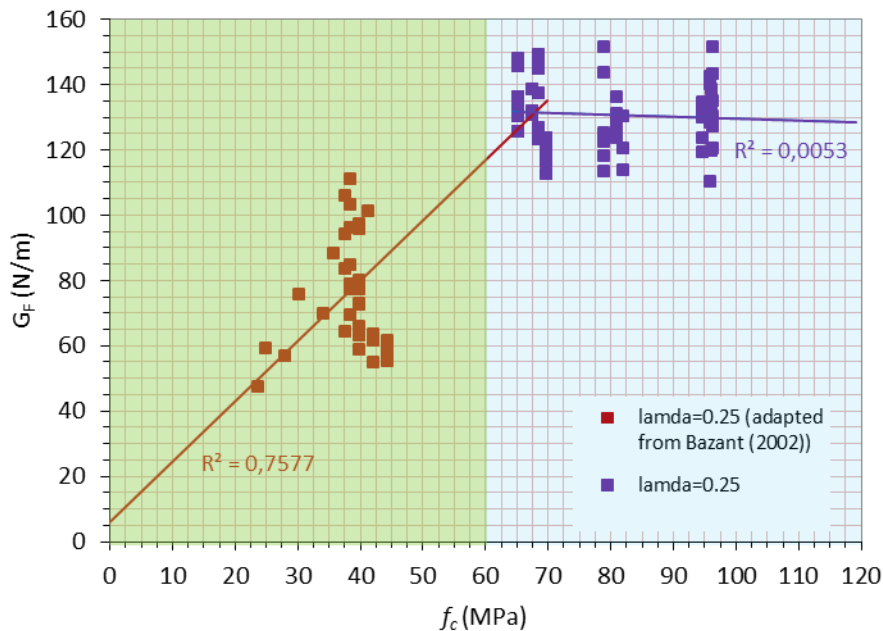
427 concrete, can play an important role in distributing stress in the material and hence

428 influence the initiation of cracks. Therefore, w/b ratio and aggregate grading are

429 factors influencing the quality of the interfacial zone. Consequently, the reduction  
 430 of w/b ratio of concrete from 0.30 to 0.20 increases  $K_{IC}$  by  $\approx 26\%$  and  $f_c$  by a  
 431 similar amount,  $\approx 23\%$ . At the same time, the measured values of  $G_F$  were found  
 432 to essentially independent of w/b within the accuracy of the measurements. This is  
 433 thought to be because, as elaborated in Section 5.1, the aggregate grading affects  
 434 the dissipated fracture energy of concrete. Thus,  $G_F$  is dependent on the crack path  
 435 and is influenced primarily by the quality of aggregate, which was kept constant,  
 436 and the aggregate size distribution, which was varied.

437

438 The relationship between the measured fracture energy,  $G_F$ , and  $f_c$ , for  
 439 compressive strength more than 60 MPa is shown in Figure 8, along with  
 440 published data taken from Bazant and Becq-Giraudon (2002) for normal strength  
 441 concrete ( $f_c < 50$  MPa) with the same value of  $\lambda (= 0.25)$ .



442

443 **Fig. 8** Effect of compressive strength on the  $G_F$  of concrete

444

445 Although the experimental data shows some scatter, it can be seen that initially  $G_F$   
446 increases linearly, and a regression on the trend line yields an  $f_c$ ,  $R^2 = 0.758$ . This  
447 relationship holds where the concrete strength is below  $\approx 60$ -70 MPa. However,  
448 where the strength of the concrete is above this value,  $G_F$ , remains at around 130  
449 ( $\pm 20$ ) N/m and appears unaffected by any increase in  $f_c$ , indeed a regression on  
450 the trend line yields an  $R^2 = 0.005$  which suggests that the two variables are not  
451 correlated. Given that these results are for concrete manufactured with the same  
452 value of  $\lambda$ , this behaviour is not related to any change in the coarseness of internal  
453 random grain structure. Thus, for the normal strength concrete mixes ( $f_c < 60 \pm 5$   
454 MPa), where the fracture path is dominated by aggregate cement de-bonding (a  
455 more tortuous crack path exists),  $G_F$  increases with  $f_c$  (and hence w/b ratio) due to  
456 the improved strength of both the aggregate/cement bond and the cement matrix  
457 itself. When  $f_c$  exceeds  $\approx 60 \pm 5$  MPa then the aggregate/cement bond strength is  
458 sufficient that the fracture path can pass through the aggregate particles. When  
459 this transition occurs the fracture path becomes dominated by aggregate failure  
460 and as a consequence  $G_F$  is limited by the strength of the aggregate despite the  
461 compressive strength of the concrete,  $f_c$ , continuing to increase. Therefore, the  
462 maximum value of  $G_F$  for a given high strength concrete will be limited by the  
463 strength of the aggregate.

464

## 465 **6 Conclusions**

466 The aim of this paper was to investigate the effect of aggregate size distribution  
467 (induced by aggregate grading) on the fracture behaviour of high-strength  
468 concrete. Based on a qualitative analysis of the post-peak softening curve

469 behaviour and a quantitative analysis of the measured fracture energy ( $G_F$ ) and  
470 critical stress intensity factor ( $K_{IC}$ ) the following conclusion can be drawn:

- 471 1. The compressive strength and critical stress intensity factor of the high-  
472 strength concrete demonstrated the expected dependence on the w/b ratio,  
473 i.e. increase in w/b ratio reduces both the compressive strength and the  
474 critical stress intensity factor.
- 475 2. The size distribution of the aggregate, measured using the coarseness of  
476 internal random grain structure of the concrete,  $\lambda$ , has a significant  
477 influence on the fracture energy of high strength concrete. Changing  $\lambda$   
478 from 0.32 to 0.22 resulted in a 50% increase in measured values of  $G_F$ .
- 479 3. The critical stress intensity factor,  $K_{IC}$ , is relatively insensitive to the  
480 aggregate size distribution; however, higher proportion of coarser grain  
481 size tends to increase the  $K_{IC}$ .
- 482 4. On the basis of the measured values of  $G_F$  and the  $K_{IC}$ , the aggregate size  
483 distribution and w/b ratio were found to influence the level of ductility  
484 exhibited by high strength concrete.
- 485 5. The maximum value of  $G_F$  for a high strength concrete will be limited by  
486 the strength of the aggregate.

487 The fracture parameters obtained in these experiments enrich the factors which  
488 influence the fracture behaviour of high-strength concrete and can be used to  
489 further refine predicted fracture energy formula used in developing design criteria.

490

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494

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